

# Hierarchical SnO<sub>2</sub> Nanoflakes Integrated with Carbon Nanofibers as an Advanced Anode Material for High-Performance Lithium-Ion Batteries

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(Received for review July 4, 2024; Revision received August 8, 2024; Accepted August 9, 2024)

## Abstract

Lithium-ion batteries (LIBs) have attracted significant attention as potential energy storage solutions due to their high energy density, minimal self-discharge, extended cycle life, and absence of memory effects. However, conventional LIBs use graphite as the anode material and as a result struggle to meet the increasing demand for higher energy density because of the low theoretical capacity of graphite. In order to enhance Li storage capacity and address the current limitations of LIBs, this study designed and analyzed SnO<sub>2</sub> nanoflakes/CNF, which is an advanced anode material with a unique hierarchical structure synthesized via a facile method involving incipient wetness followed by annealing. The *in-situ* formed SnO<sub>2</sub> nanoflakes improve the electrolyte accessibility and shorten the ion and electron transport pathways, thereby enhancing the reaction kinetics. Additionally, the CNF matrix enhances the electrical conductivity, accelerates electron transport, and mitigates volume changes. The integrated SnO<sub>2</sub> nanoflakes/CNF cell demonstrated outstanding cycling performance and excellent rate capability, achieving a notable reversible capacity of 636 mAh g<sup>-1</sup> after 100 cycles at 0.1 C. This study provides valuable insights into the design of high-efficiency anode materials for the advancement of high-performance LIBs.

**Keywords** : Hierarchical structure, SnO<sub>2</sub> nanoflakes, Carbon nanofiber, Anode, Rechargeable Li-SnO<sub>2</sub> batteries

## 1. Introduction

Lithium-ion batteries (LIBs) with high energy density and extended lifespans are essential for addressing the increasing energy storage requirements of electric vehicles, smart grids, and other energy storage systems[1-3]. Graphite, the standard anode material in LIBs, struggles to meet the rising demand for higher energy density due to its low theoretical capacity (372 mAh g<sup>-1</sup>)[4,5]. Thus, developing advanced anode materials with higher theoretical capacities is essential to surpass the current limitations of LIBs. Transition-metal oxides (TMOs) have emerged as promising alternatives to graphite owing to their rich in natural resources

and outstanding Li storage capacity[6-8]. Among TMOs, SnO<sub>2</sub> is notable as an anode material for lithium storage due to its high theoretical capacity, non-toxicity, and cost-effectiveness. However, SnO<sub>2</sub> undergoes significant volumetric changes (>300%) during charge-discharge cycles, leading to rapid capacity degradation, poor cycling performance, and inferior rate capability[9,10].

To tackle these challenges, a synergistic approach that integrates tailored SnO<sub>2</sub> nanoparticles with carbon matrices has been shown to effectively enhance the electrochemical performance of rechargeable Li-SnO<sub>2</sub> batteries. The tailored SnO<sub>2</sub> nanoparticles, with their unique structure, can shorten ion transport pathways and improve electrolyte accessibility, thereby facilitating reaction

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<https://doi.org/10.7464/ksct.2024.30.3.267> pISSN 1598-9712 eISSN 2288-0690

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kinetics. The carbon matrices, characterized by high electrical conductivity, enhance electron transport and buffer volume changes, further improving electrochemical performance. For example, Lu et al.[11] reported three-dimensional reduced graphene oxide-encapsulated SnO<sub>2</sub> nanoparticles, achieving a remarkable capacity retention of 1,592 mAh g<sup>-1</sup> over 500 cycles at 0.5 A g<sup>-1</sup>. Similarly, Wang et al.[12] synthesized a Sn/SnO<sub>2</sub>/C nanohybrid, where ultrafine SnO<sub>2</sub>/Sn nanoparticles encapsulated within an adjustable mesoporous carbon matrix maintained a high-capacity retention of 1,105 mAh g<sup>-1</sup> at 200 mA g<sup>-1</sup> after 290 cycles. Undoubtedly, the rational design of SnO<sub>2</sub> with a unique structure integrated with a carbon matrix can significantly enhance the cycling performance and rate capability of rechargeable Li-SnO<sub>2</sub> batteries owing to the synergistic effect between the components.

Hence, we have elaborately designed a hierarchical structure in which SnO<sub>2</sub> nanoflakes are integrated within carbon nanofibers (CNF) using a facile incipient wetness technique followed by an annealing process, effectively address challenges associated with rechargeable Li-SnO<sub>2</sub> batteries. The CNF matrix plays a significant role in increasing conductivity, accelerating electron transport, and alleviating volume changes. Meanwhile, the *in-situ* formed SnO<sub>2</sub> nanoflakes reduce pathways for ion and electron transport, thereby enhancing reaction kinetics. Consequently, the unique hierarchical structure of the SnO<sub>2</sub> nanoflakes/CNF anode material demonstrates exceptional cycling performance and rate capability, achieving a remarkable lithiation capacity retention of 636 mAh g<sup>-1</sup> at 0.1 C after 100 cycles. This hierarchical SnO<sub>2</sub> nanoflakes/CNF composite shows outstanding rate performance and cycling stability, making it an excellent candidate for advanced LIB anodes.

## 2. Materials and methods

### 2.1 Materials

Polyacrylonitrile (PAN, average MW 150,000, Sigma-Aldrich), N, N-dimethylformamide (DMF, 99.0%, Samchun Pure Chemical Co., Ltd.), tin (II) chloride dihydrate (SnCl<sub>2</sub>·2H<sub>2</sub>O, 98.0%, Sigma-Aldrich), Tin oxalate (Sn(C<sub>2</sub>O<sub>4</sub>), 98.0%, Sigma-Aldrich), hexamethylenetetramine (HMTA, 99.0%, Samchun Pure Chemical Co., Ltd.), thiourea (SCN<sub>2</sub>H<sub>4</sub>, 98.0%, Samchun Pure Chemical Co., Ltd.), ethylene glycol (99.8%, Sigma-Aldrich), Tin oxide nanopowder (Sigma-Aldrich), and N-methylpyrrolidone (NMP, 99.5%, Samchun Pure Chemical Co., Ltd.) were used as received.

### 2.2 Preparation of SnO<sub>2</sub> nanoflakes/CNF and SnO<sub>2</sub> nanoflakes composites

To synthesize SnO<sub>2</sub> nanoflakes/CNF composite, a series of steps were followed, beginning with the preparation of CNF using

a straightforward electrospinning method. Initially, PAN was dissolved in DMF at 60 °C for 12 h with continuous stirring. The PAN solution was then loaded into a syringe fitted with a 21-gauge metallic needle. Electrospinning was performed by applying a high voltage of 17 kV at a flow rate of 3 mL h<sup>-1</sup>, with the collector placed 18 cm from the needle. The resulting PAN fibers were stabilized at 300 °C for 3 h in an air atmosphere and then carbonized at 800 °C for 5 h with a heating rate of 3 °C min<sup>-1</sup> under an Ar atmosphere. Simultaneously, a solution of Sn(C<sub>2</sub>O<sub>4</sub>) and HMTA in an equal molar ratio was prepared by dissolving the compounds in 500 mL of distilled water to achieve a concentration of approximately 0.05 M, forming a white milky solution. The CNF was immersed in this solution and heated at 150 °C for 1 h. After washing, the mixture was calcinated at 350 °C for 3 h in an air atmosphere, resulting in the formation of SnO<sub>2</sub> nanoflakes/CNF composite.

For a thorough comparison of electrochemical performance, SnO<sub>2</sub> nanoflakes were also synthesized according to a previously reported method[13]. A 0.05 M solution of SnCl<sub>2</sub>·2H<sub>2</sub>O and a 0.1 M solution of thiourea were added to 100 mL of ethylene glycol in a round-bottom flask and heated at 170 °C for 1 h. After washing, the mixture was calcinated at 500 °C for 3 h in an air atmosphere, resulting in the formation of SnO<sub>2</sub> nanoflakes.

### 2.3 Characterization of materials

The external morphologies of the synthesized samples were observed using field-emission scanning electron microscopy (FE-SEM, TESCAN, MIRA3 LM), while their internal structures were investigated by transmission electron microscopy (TEM, FEI, TF30ST). The crystal structures of the samples were examined using X-ray diffraction (XRD, Bruker, D2 Phaser) with Cu K $\alpha$  radiation over a range of 20 ~ 80°. The specific surface area was determined through Brunauer-Emmet-Teller analysis (BET, Micromeritics, ASAP 2010). To quantify the SnO<sub>2</sub> content in the SnO<sub>2</sub> nanoflakes/CNF composite, thermogravimetric analysis (TGA, TA Instruments, Q50) was performed from room temperature to 650 °C at a heating rate of 10 °C min<sup>-1</sup> in an O<sub>2</sub> atmosphere.

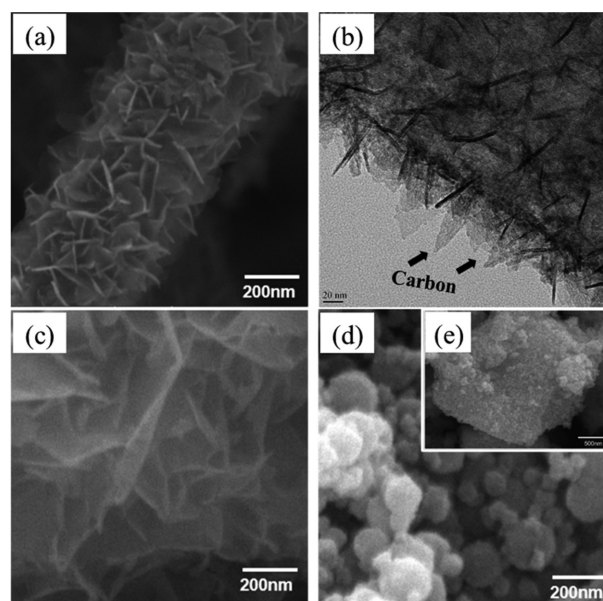
### 2.4 Electrochemical characterization

The anode electrodes were fabricated by mixing 80% active material (SnO<sub>2</sub> nanoflakes/CNF, SnO<sub>2</sub> nanoflakes, SnO<sub>2</sub> nanoparticles), 10% Super P, and 10% poly(vinylidene fluoride) (PVDF) in N-methyl-2-pyrrolidone (NMP). The homogeneous slurry was then uniformly coated onto copper foil, dried at 80 °C for 12 h under vacuum conditions, and subsequently punched into discs with a diameter of 10 mm. Lithium metal foil served as the counter electrode, Celgard<sup>®</sup> 2400 as the separator, and a 1 M lithium hexafluorophosphate (LiPF<sub>6</sub>) dissolved in a mixture

of ethylene carbonate (EC) and diethylene carbonate (DEC) with a 1:1 volume ratio as the electrolyte. Galvanostatic charge/discharge testing was performed at varying current densities within a voltage range of 0.01 ~ 3.0 V using a WBCS3000 battery cycler (WonA Tech. Co., Ltd., South Korea). Cyclic voltammetry (CV) measurements were conducted at a scan rate of 0.5 mV s<sup>-1</sup>. Electrochemical impedance spectroscopy (EIS) was carried out with an impedance analyzer (ZIVE SP2, WonA Tech. Co., Ltd.) over a frequency range of 0.1 Hz to 100 kHz.

### 3. Results and discussion

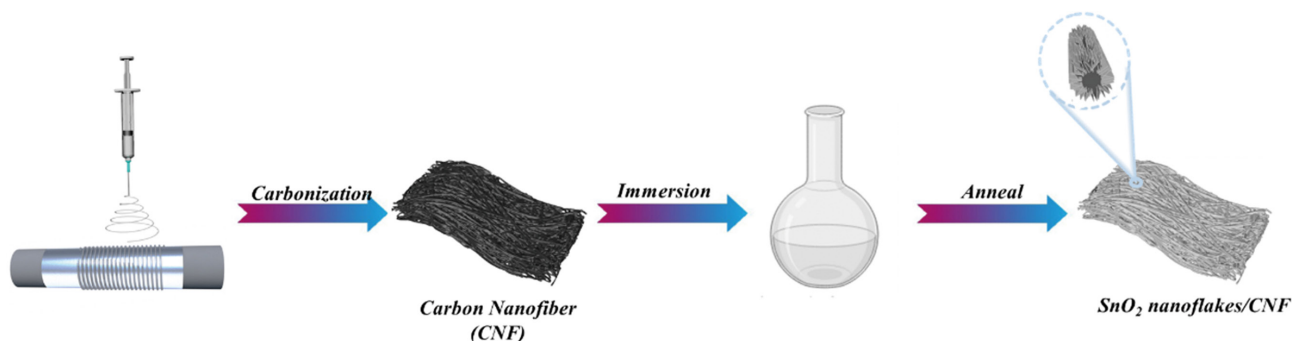
Figure 1 illustrates the preparation process of SnO<sub>2</sub> nanoflakes on CNF. Through a facile annealing procedure, interlaced SnO<sub>2</sub> nanoflakes were uniformly grown on the CNF. The resulting SnO<sub>2</sub> nanoflakes/CNF composite exhibits a well-dispersed fibrous structure with widths ranging from 400 to 500 nm, as depicted in Figure 2a. Moreover, the interlaced SnO<sub>2</sub> nanoflakes possess a thickness of approximately 10 nm. Further investigation of the microstructure of the SnO<sub>2</sub> nanoflakes/CNF was conducted using TEM analysis, shown in Figure 2b. The TEM image confirms that the interlaced SnO<sub>2</sub> nanoflakes are densely anchored on the CNF, consistent with the SEM observation. Additionally, the TEM image reveals the presence of prominent carbon within the interstices of the interlaced SnO<sub>2</sub> nanoflakes. This prominent carbon substantially enhances electrical conductivity and effectively buffers the volume changes during the lithiation and delithiation processes. In comparison, the morphologies of the synthesized SnO<sub>2</sub> nanoflakes and commercial SnO<sub>2</sub> nanoparticles are depicted in Figure 2(c ~ e). The SnO<sub>2</sub> nanoflakes exhibit similar interlaced nanoflake morphological characteristics to those of SnO<sub>2</sub> nanoflakes/CNF, as shown in Figure 2c. In contrast, individual SnO<sub>2</sub> nanoparticles display sphericity with diameters ranging from 10 to 100 nm (Figure 2d), and these nanoparticles tend to agglomerate into a bulk morphology (Figure 2e). The surface morphological and internal structural analysis reveal that the synthesized SnO<sub>2</sub> nanoflakes/CNF



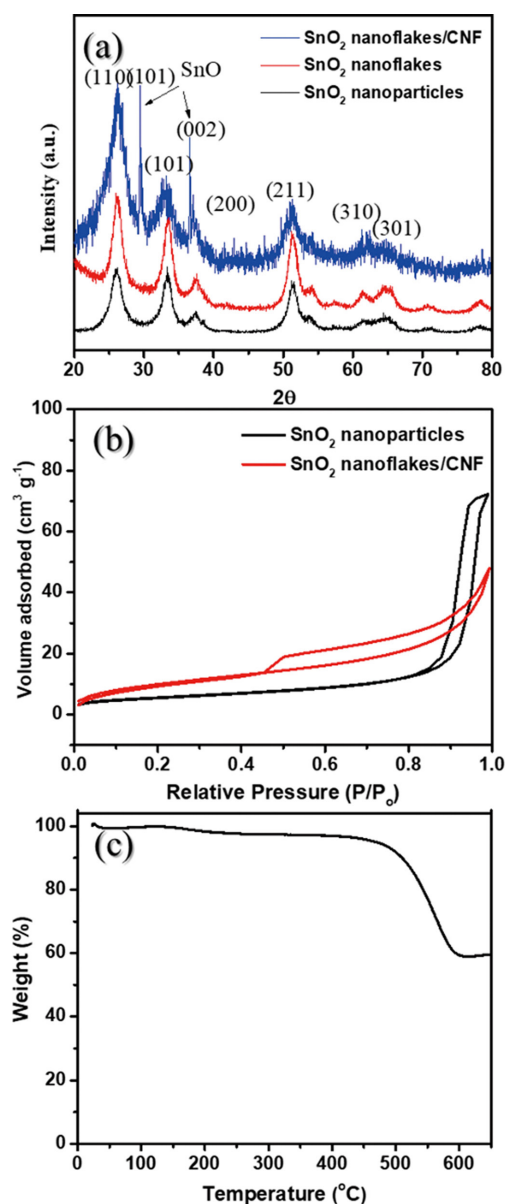
**Figure 2.** (a) FE-SEM image of SnO<sub>2</sub> nanoflakes/CNF, (b) TEM image of SnO<sub>2</sub> nanoflakes/CNF, FE-SEM images of (c) SnO<sub>2</sub> nanoflakes, and (d ~ e) SnO<sub>2</sub> nanoparticles.

composite features a unique hierarchical structure. The CNF acts as a supporting matrix, greatly improving electrical conductivity and preventing the agglomeration of the SnO<sub>2</sub> nanoflakes generated in situ. Additionally, the interlaced SnO<sub>2</sub> nanoflakes formed in situ enhance electrolyte accessibility and increase active sites, accelerating the reaction kinetics.

The XRD analysis presented in Figure 3a facilitated the identification of the phases in SnO<sub>2</sub> nanoflakes/CNF, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoparticles. The main diffraction peaks at 26.6°, 33.2°, 37.2°, 50.7°, 60.6°, and 63.5° for SnO<sub>2</sub> nanoflakes/CNF corresponded precisely to the (110), (101), (200), (211), (310), and (301) crystal planes, respectively, which showed a high match with SnO<sub>2</sub> nanoflakes and SnO<sub>2</sub> nanoparticles, affirming the successful synthesis of SnO<sub>2</sub> in the SnO<sub>2</sub> nanoflakes/CNF composite[14,15]. Additionally, two distinct peaks at 29.1° and 35.7° were observed, corresponding to the (101) and (002) planes of SnO (JCPDS No. 06-0395). The presence of SnO is likely attributable to the thermal

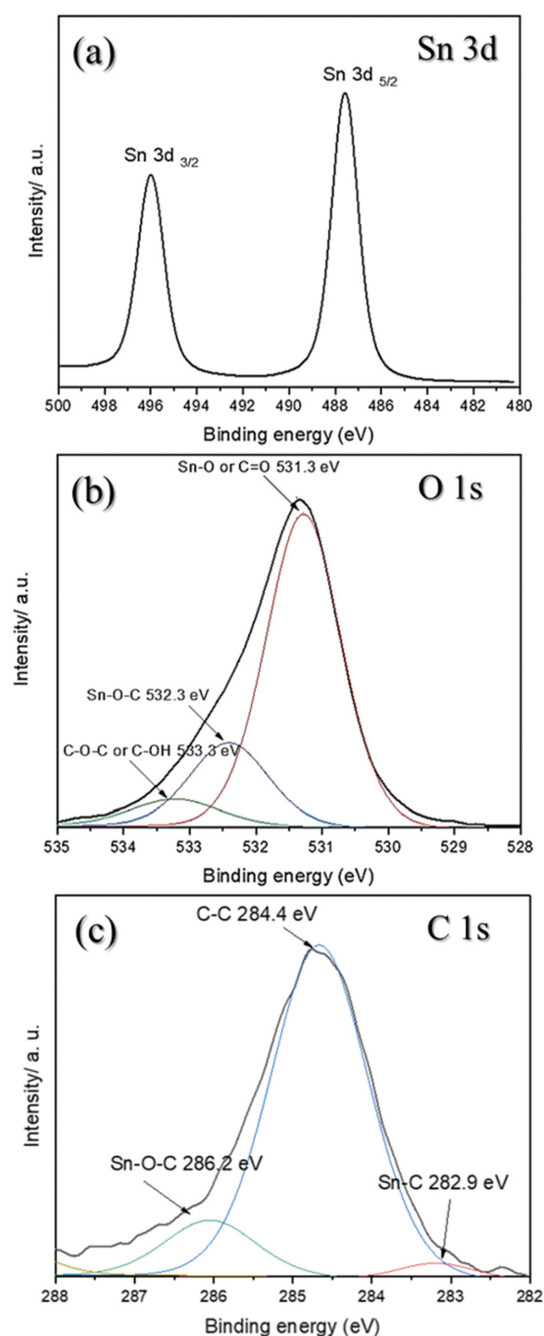


**Figure 1.** Illustration of the sequential fabrication steps for SnO<sub>2</sub> nanoflakes/CNF composite.



**Figure 3.** (a) XRD patterns of SnO<sub>2</sub> nanoparticles, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoflakes/CNF, (b) nitrogen adsorption-desorption isotherms of SnO<sub>2</sub> nanoparticles, and SnO<sub>2</sub> nanoflakes/CNF, (c) TGA plot of SnO<sub>2</sub> nanoflakes/CNF composite.

decomposition of SnC<sub>2</sub>O<sub>4</sub>[16]. The N<sub>2</sub> adsorption/desorption isotherms for SnO<sub>2</sub> nanoflakes/CNF and SnO<sub>2</sub> nanoparticles are illustrated in Figure 3b. The SnO<sub>2</sub> nanoflakes/CNF displayed a typical type IV isotherm with an H1-type hysteresis loop. Conversely, the SnO<sub>2</sub> nanoparticles exhibited a combination of type II and type IV isotherms with an H3-type hysteresis loop. The surface area of SnO<sub>2</sub> nanoflakes/CNF (40 m<sup>2</sup> g<sup>-1</sup>) was larger than that of the SnO<sub>2</sub> nanoparticles (18 m<sup>2</sup> g<sup>-1</sup>), indicating that SnO<sub>2</sub> nanoflakes/CNF is more conducive to reactions due to enhanced electrolyte accessibility. To accurately determine the SnO<sub>2</sub> content in the SnO<sub>2</sub> nanoflakes/CNF composite, TGA was conducted in



**Figure 4.** High-resolution (a) Sn 3d, (b) O 1s, and (c) C 1s XPS data of SnO<sub>2</sub> nanoflakes/CNF, respectively.

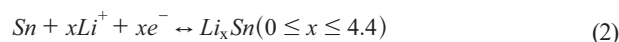
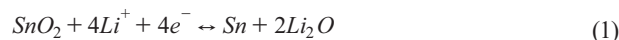
an O<sub>2</sub> atmosphere from room temperature to 650 °C at a heating rate of 10 °C min<sup>-1</sup> (Figure 3c). A substantial mass loss occurred between 450 and 550 °C, attributed to the oxidation of carbon within the SnO<sub>2</sub> nanoflakes/CNF composite[17]. Consequently, the SnO<sub>2</sub> content in the composite was estimated to be approximately 60 wt%.

The chemical properties of the SnO<sub>2</sub> nanoflakes/CNF composite were examined through XPS analysis (Figure 4a ~ c). The high-resolution Sn 3d spectrum of the SnO<sub>2</sub> nanoflakes/CNF composite reveals distinct peaks at 487.6 eV and 496.1 eV, corresponding to

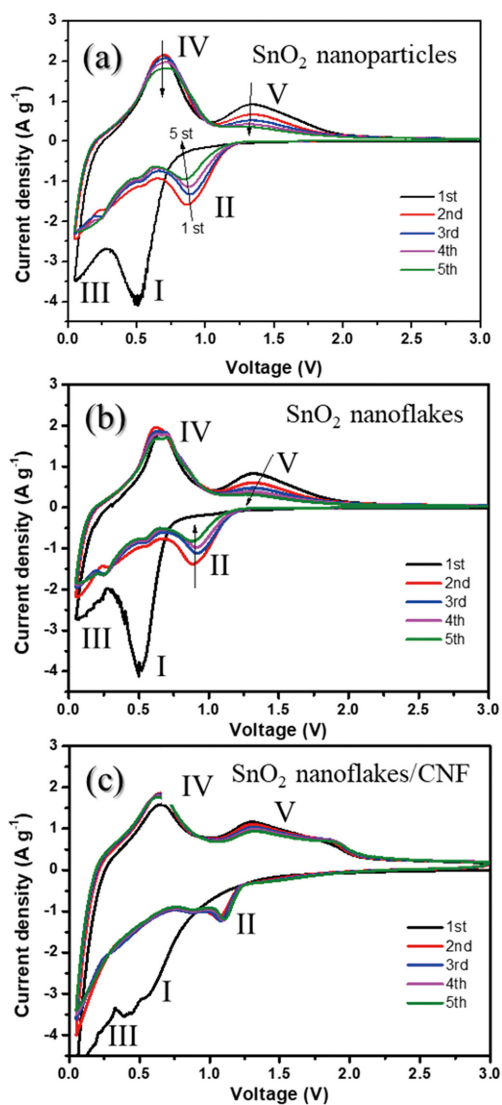
Sn 3d<sub>5/2</sub> and Sn 3d<sub>3/2</sub>, respectively. The O 1s XPS spectrum for the composite shows three peaks at 531.3 eV, 532.3 eV, and 533.3 eV, attributable to Sn-O/C=O, Sn-O-C, and C-O-C/C-OH, respectively [18-20]. Furthermore, the C 1s spectrum exhibits peaks at 282.9 eV and 286.2 eV, indicative of the formation of Sn-C and Sn-O-C bonds[21-23]. This evidence points to robust chemical bonding at the interface between SnO<sub>2</sub> and CNF. The presence of Sn-C and Sn-O-C interfacial bonds not only firmly anchors the SnO<sub>2</sub> nanoflakes to the CNF matrix but also significantly enhances the interaction between the two components[24].

The reaction mechanisms of SnO<sub>2</sub> nanoflakes/CNF, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoparticles cells are investigated using the CV curves depicted in Figure 5(a ~ c). SnO<sub>2</sub> nanoflakes and SnO<sub>2</sub> nanoparticles cells exhibit similar electrochemical behaviors. During the initial cathodic scan, a voltage hysteresis around 0.5 V

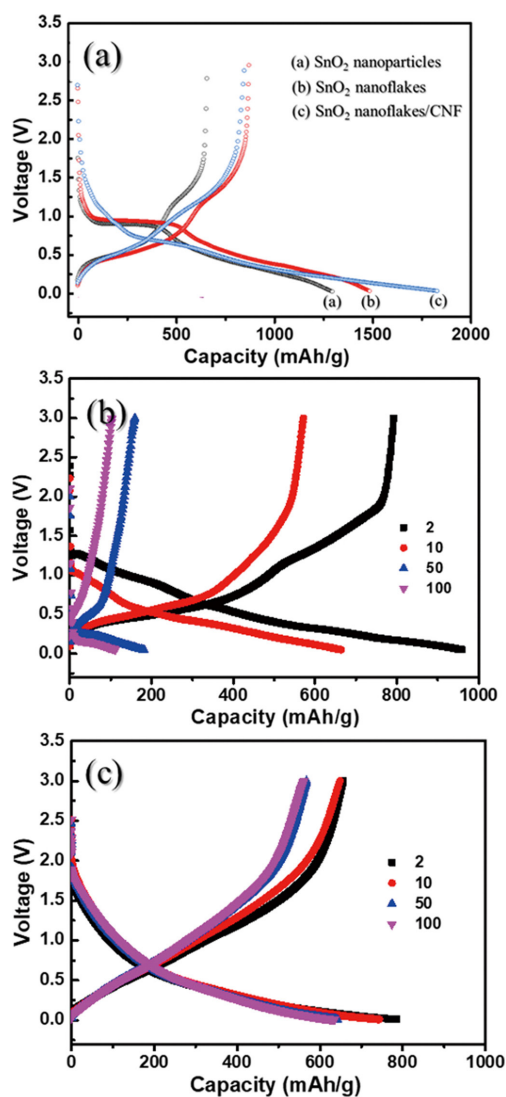
(Peak I) is observed, which is associated with the formation of a solid electrolyte interface (SEI) film and the irreversible decomposition of the electrolyte[25,26]. Interestingly, the current values at Peak I are significantly higher for the SnO<sub>2</sub> nanoflakes and SnO<sub>2</sub> nanoparticles cells compared to SnO<sub>2</sub> nanoflakes/CNF, indicating more extensive irreversible reactions in the former. This suggests that the well-engineered structure of SnO<sub>2</sub> nanoflakes/CNF effectively minimizes irreversible reactions during the initial lithiation process, thereby enhancing initial Coulombic efficiency and active material utilization[27]. In subsequent cycles, the lithiation voltage from Peak I shifts to Peak II, corresponding to the reduction of SnO<sub>2</sub> to Sn (Equation (1)). Another lithiation voltage (Peak III) is related to the alloying reaction from Sn to Li-Sn alloy (Equation (2)). The anodic peaks (Peak IV and Peak V) indicate the oxidation of Li-Sn back to Sn, and further oxidation to SnO<sub>2</sub>[28,29]. The current value at Peak III for SnO<sub>2</sub> nanoflakes/CNF exceeds that of SnO<sub>2</sub> nanoflakes and SnO<sub>2</sub> nanoparticles, suggesting a more complete alloying reaction. Additionally, Peaks II, III, IV, and V for SnO<sub>2</sub> nanoflakes/CNF show consistent repeatability in the CV curves, reflecting a stable reaction environment. In contrast, the same peaks for SnO<sub>2</sub> nanoflakes and SnO<sub>2</sub> nanoparticles cells exhibit a continuous decline, indicating an unstable reaction environment. This stability highlights the advantages of the SnO<sub>2</sub> nanoflakes/CNF structure in ensuring reliable and reproducible electrochemical processes.



The initial charge-discharge profiles of SnO<sub>2</sub> nanoflakes/CNF, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoparticles cells at 0.1 C are shown in Figure 6a. The SnO<sub>2</sub> nanoflakes/CNF composite exhibited the highest lithiation capacity of 1,800 mAh g<sup>-1</sup>, followed by SnO<sub>2</sub> nanoflakes of 1,470 mAh g<sup>-1</sup>, and SnO<sub>2</sub> nanoparticles of 1,270 mAh g<sup>-1</sup>. This indicates that the SnO<sub>2</sub> in the SnO<sub>2</sub> nanoflakes/CNF composite demonstrates the highest utilization, attributed to the synergistic effect between SnO<sub>2</sub> nanoflakes and CNF. In the subsequent charge-discharge profiles of SnO<sub>2</sub> nanoflakes (Figure 6b), capacity degradation occurred due to severe volume changes during the lithiation-delithiation processes. Conversely, from the 2nd cycle onward, the charge-discharge profiles of the SnO<sub>2</sub> nanoflakes/CNF cell (Figure 6c) maintained excellent repeatability, indicating that the SnO<sub>2</sub> nanoflakes/CNF cell possesses superior repeatability and cyclability. The cycling performance of SnO<sub>2</sub> nanoflakes/CNF, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoparticles cells is shown in Figure 7a. After 100 cycles at 0.1 C, the SnO<sub>2</sub> nanoflakes/CNF cell retained a higher capacity of approximately 636 mAh g<sup>-1</sup>,

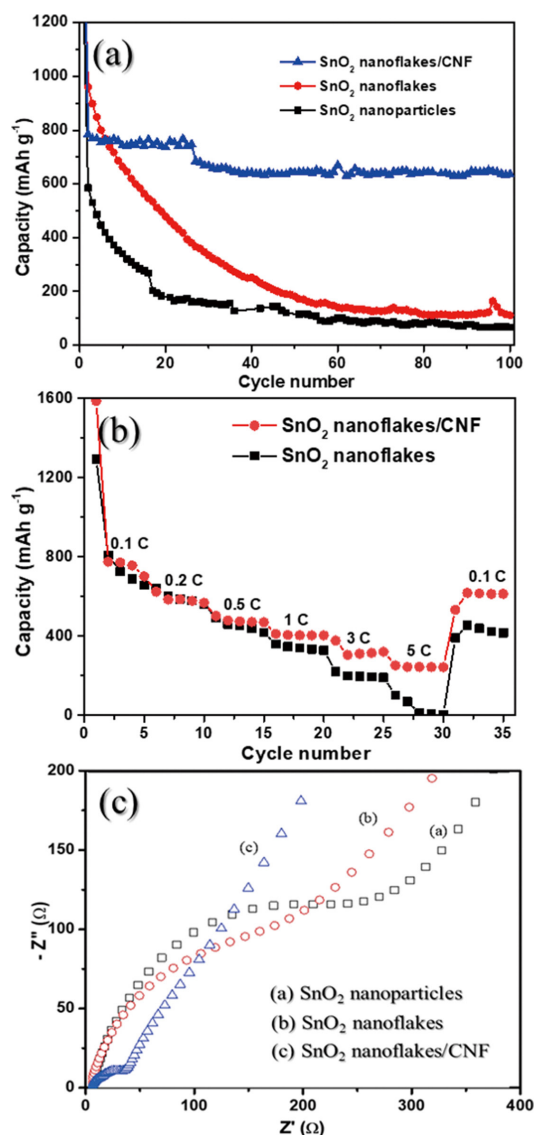


**Figure 5.** CV curves of (a) SnO<sub>2</sub> nanoparticles, (b) SnO<sub>2</sub> nanoflakes, (c) SnO<sub>2</sub> nanoflakes/CNF (scan rate: 0.5 mV s<sup>-1</sup>; 0.01 ~ 3.0 V).



**Figure 6.** (a) initial charge–discharge profiles of SnO<sub>2</sub> nanoparticles, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoflakes/CNF, subsequent charge-discharge profiles of (b) SnO<sub>2</sub> nanoflakes, (c) SnO<sub>2</sub> nanoflakes/CNF at 0.1 C.

corresponding to 81.1% of the capacity from the 2nd cycle. The electrochemical performance of SnO<sub>2</sub> nanoflakes/CNF exceeds the previously reported electrochemical performance of SnO<sub>2</sub>/carbon composites, as shown in Table 1[30-34]. In contrast, the SnO<sub>2</sub>



**Figure 7.** (a) cycling performances of SnO<sub>2</sub> nanoparticles, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoflakes/CNF at 0.1 C, (b) rate performances of SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoflakes/CNF, and (c) EIS plots of SnO<sub>2</sub> nanoparticles, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoflakes/CNF after 100 cycles.

nanoflakes and SnO<sub>2</sub> nanoparticles cells exhibited significantly lower retained capacities of only 111 and 67 mAh g<sup>-1</sup>, respectively,

**Table 1.** Summary of previous studies on SnO<sub>2</sub>/carbon composite-based anode for LIBs

Anode material employed	Current density (mA g <sup>-1</sup> )	Initial lithiation capacity (mAh g <sup>-1</sup> )	Retained lithiation capacity (mAh g <sup>-1</sup> )	Cycles	Refs
Flexible SnO <sub>2</sub> /G-CNT paper	100	990	387	50	30
SnO <sub>2</sub> /G-CNT	100	1,050	502	50	31
SnO <sub>2</sub> /G-CNT	200	1,806	842	40	32
SnO <sub>2</sub> -G	50	2,140	649	30	33
SnO <sub>2</sub> /gC	50	1,675	592	100	34
SnO <sub>2</sub> /MWNT	50	907	231	50	
SnO <sub>2</sub> nanoflakes/CNF	0.1 C	1,800	636	100	This work

after 100 cycles. This highlights the significant improvement in the reversibility of the conversion reaction provided by the SnO<sub>2</sub> nanoflakes/CNF composite. Moreover, a rate capability test was conducted for the SnO<sub>2</sub> nanoflakes/CNF and SnO<sub>2</sub> nanoflakes cells, as shown in Figure 7b. The test began at 0.1 C for 5 cycles, progressively increased to 5 C, and then decreased back to 0.1 C. The SnO<sub>2</sub> nanoflakes/CNF cell achieved initial lithiation capacities of 1,587, 625, 503, 412, 378, 252, and 614 mAh g<sup>-1</sup>, while the SnO<sub>2</sub> nanoflakes cell achieved 1,295, 641, 493, 360, 221, 102, and 417 mAh g<sup>-1</sup> at 0.1, 0.2, 0.5, 1, 3, 5, and 0.1 C, respectively. The superior rate capability of the SnO<sub>2</sub> nanoflakes/CNF composite can be attributed to its unique hybrid structure, where the CNF enhances electrical conductivity and buffers volume changes during charge-discharge processes. Additionally, the *in-situ* generated SnO<sub>2</sub> nanoflakes shorten the ion transport path and increase the contact area between the active material and the electrolyte, thereby enhancing the electrochemical performance. Furthermore, the EIS spectra for SnO<sub>2</sub> nanoflakes/CNF, SnO<sub>2</sub> nanoflakes, and SnO<sub>2</sub> nanoparticles cells were obtained in the fully delithiated state after 100 cycles (Figure 7c). The Nyquist plots for these cells display a depressed semicircle in the high-middle frequency region, indicative of the charge transfer resistance ( $R_{ct}$ ), and a sloped line in the low-frequency region, representing the Warburg impedance ( $R_w$ ) associated with Li<sup>+</sup> ion diffusion[35,36]. The  $R_{ct}$  of the cycled cell with SnO<sub>2</sub> nanoflakes/CNF is the lowest at 53  $\Omega$ , compared to 379  $\Omega$  for the cycled cell with SnO<sub>2</sub> nanoflakes and 414  $\Omega$  for the cell with SnO<sub>2</sub> nanoparticles. This is evidenced by the smaller depressed semicircle in the Nyquist plot for the SnO<sub>2</sub> nanoflakes/CNF cell, indicating faster reaction kinetics. The CNF serves as a conductive matrix, significantly enhancing electrical conductivity and reducing resistance in the SnO<sub>2</sub> nanoflakes/CNF cell. The Warburg factor ( $\sigma$ ) is calculated from the slope of the plot of the real part of resistance against the inverse square root of the angular frequency in the low-frequency range, using Equation (3):

$$D = \frac{R^2 T^2}{2A^2 n^2 F^4 C^2 \sigma^2} \quad (3)$$

where  $R$  is the gas constant,  $T$  is the absolute temperature,  $A$  is the electrode surface area,  $F$  is the Faraday constant, and  $C$  is the molar concentration of Li ions in the active material. The lithium diffusion coefficient ( $D$ ) for the cycled SnO<sub>2</sub> nanoflakes/CNF cell is calculated to be  $4.3 \times 10^{-11}$  cm<sup>2</sup> s<sup>-1</sup>, confirming the rapid reaction kinetics in the SnO<sub>2</sub> nanoflakes/CNF cell.

#### 4. Conclusions

A hierarchical SnO<sub>2</sub> nanoflakes/CNF composite was effectively

synthesized using a simple incipient wetness method followed by annealing. The composite exhibits outstanding cycling performance and excellent rate capability, attributed to three key factors: (1) the CNF matrix prevents the agglomeration of *in-situ* grown SnO<sub>2</sub> nanoflakes, enhances electrical conductivity, accelerates electron transport, and mitigates volume changes; (2) the *in-situ* generated SnO<sub>2</sub> nanoflakes increase active sites, enhance electrolyte accessibility and shorten ion and electron transport pathways; and (3) the synergistic interaction between SnO<sub>2</sub> nanoflakes and CNF matrix optimizes charge storage and transfer processes, improving overall electrochemical performance. As a result, the SnO<sub>2</sub> nanoflakes/CNF composite demonstrates exceptional rate capability and excellent cycling stability. This research offers valuable insights into the design of anode materials for LIBs, aiding the advancement of high-performance energy storage systems.

#### Acknowledgements

This research was supported by ‘regional innovation mega project’ program through the Korea Innovation Foundation funded by Ministry of Science and ICT (Project Number: 2023-DD-UP-0026) and the National Research Foundation of Korea (NRF) grant funded by the Korea government (MSIT) (No. RS-2023-00217581).

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